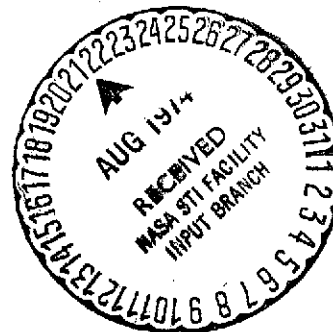


INVESTIGATION OF THE EFFECT OF CERTAIN ELEMENTS  
ON THE HIGH-TEMPERATURE STRENGTH OF ALUMINUM-TITANIUM ( $Ti_3Al$ )

O.N. Andreyev and I.I. Kornilov

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16. Abstract Elements from groups IV-VIII and IVB were used to examine the heat resistance of aluminum-titanium, when alloyed with them. It was found that when alloyed with vanadium or molybdenum, Ti <sub>3</sub> Al is more easily deformed at high temperatures. The great heat resistance of the alloy at 750° and a stress of 25 kgf/mm <sup>2</sup> occurs when alloyed with 1 technical atmosphere- % of W, Sn, Mo, Hf, Si, Zr, Tr, Nb, V.			
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According to the physicochemical theory of heat resistance, /101\* intermetallic semiconductors can be the basis of heat-resisting alloys [1]. In the Ti-Al system a number of metallic semiconductors are formed:  $Ti_3Al$ ,  $TiAl$ ,  $TiAl_3$  [2, 3]. From this series of compounds aluminum-titanium  $Ti_3Al$  has a very high heat resistance with a satisfactory technological plasticity at high temperatures [4].

In this work, a study was made of the effect of 1 technical atmosphere-% of transient metals IV-VIII and elements IVB of groups of a periodic system for the heat resistance of the  $Ti_3Al$  compound. From the IV group of metals, the following were chosen: Zr and Hf; V - V, Nb and Ta; VI - Cr, Mo and W; VII- Mn; VIII - Fe, Co and Ni; from elements of IVB group - C, Si, Ge, Sn and Pb.

Initial materials and research methods into alloys. The  $Ti_3Al$  compound was melted by a double melting method. Pure metals were used as alloying elements; zirconium and hafnium iodide, vanadium, niobium, tantalum, molybdenum, armco iron from an electron beam melt; electrolytic chromium, manganese, nickel, tin, cobalt; tungsten was introduced in the form of a Ti-W alloying element (25 wt-% W); lead, in the form of a Ti-Pb alloying element (16 wt-% Pb); carbon, in the form of a Ti-C alloying element (4 wt-% C); extremely pure silicon and germanium. Ingots weighing 500 grams were melted in a vacuum arc furnace with a nonconsumable tungsten electrode in an argon atmosphere.

The alloys were melted five times to guarantee the uniform

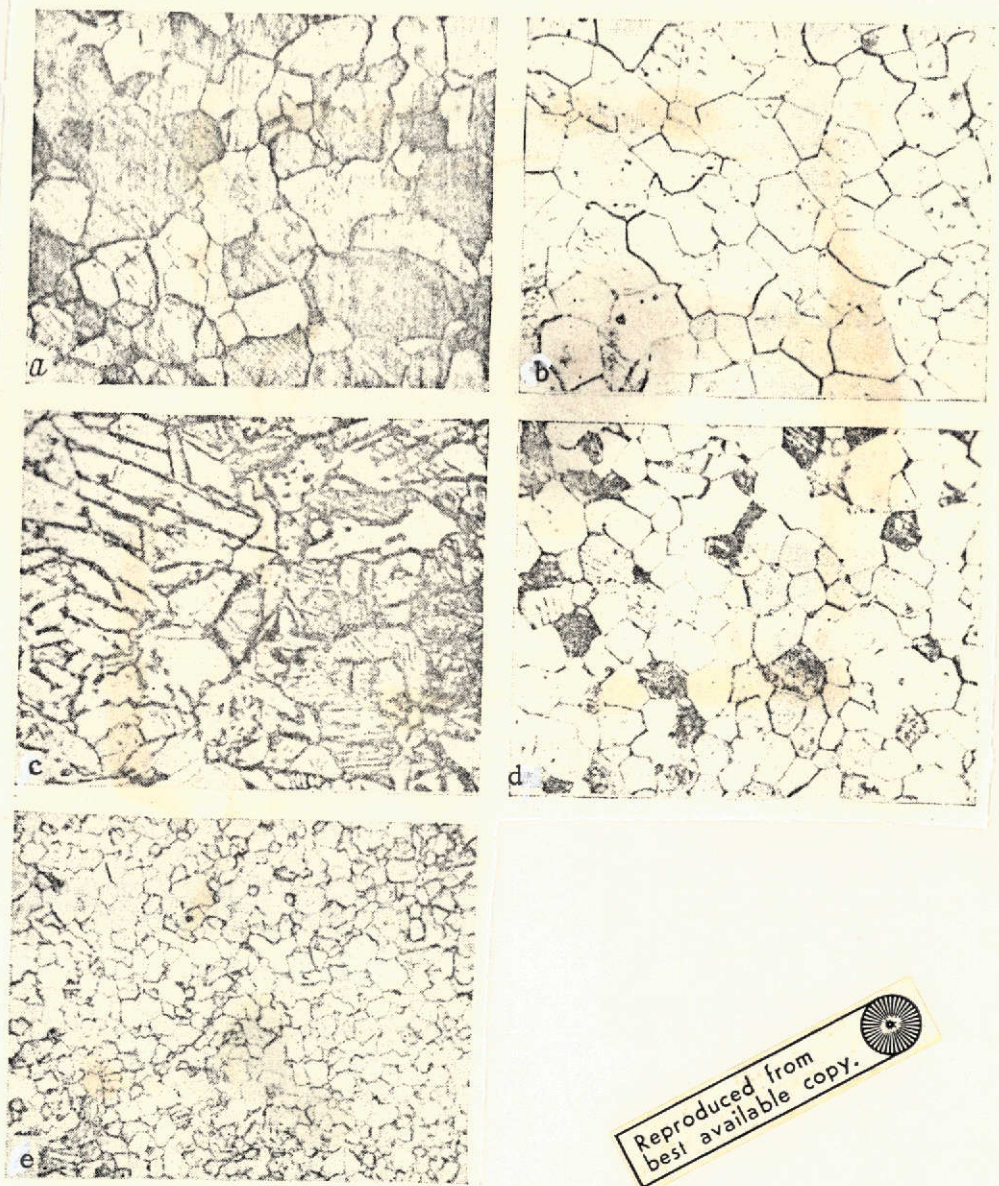
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\*Numbers in the margin indicate pagination in the foreign text.

distribution of elements. The chemical composition of the alloys was checked by relative weighing of the blend and ingot; the difference did not exceed 0.2-0.5%. The extracted ingots were cut into three sections and forged in a temperature interval of 1300-1100° into rods with a diameter of 8 mm. Alloys in this temperature range deform satisfactorily as  $\beta$ -solid solutions, with the exception of alloys containing germanium and silicon.

Alloys with a  $Ti_3Al$  base, alloyed by vanadium and molybdenum, had the greatest plasticity during deformation. The alloy specimens forged into rods underwent the following thermal treatment: 1000°, 100 hours; 800°, 100 hours; 500°, 300 hours, with subsequent slow furnace cooling to room temperature. Annealing was done in double evacuated quartz ampules, the residual pressure during welding in the ampules was  $5 \cdot 10^{-4}$  mmHg.

Apart from studying the heat resistance, the microstructure of the alloys was studied after annealing (Fig. 1). The  $Ti_3Al$  compound has a uniform structure (Fig. 1, a). The heat resistance of alloys was studied by a centrifugal bending method [1, 5] at 750° and a stress of 25 kgf/mm<sup>2</sup>. It was determined according to the time required to reach a given deflection, which was shown in work [1], and characterizes most accurately the relative heat resistance of the alloys examined. The  $Ti_3Al$  compound, alloyed by zirconium or hafnium, has a polyhedral structure of the  $\alpha_2$  solid phase on the base of this intermetallic compound. Figure 2 shows the creep curves of the alloys examined. It follows from Fig. 2, a that the alloying of Zr or Hf considerably increases the heat resistance of  $Ti_3Al$  compound, and hafnium, to a greater extent than zirconium. Whereas the time taken to reach the bending indicator of 10 mm for  $Ti_3Al$  was 3 hours, for an alloy with Zr, it was 11 hours, and for an alloy with Hf, 25 hours. Therefore, this shows that Zr and Hf increase the heat resistance of the  $Ti_3Al$  compound.



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Fig. 1. The microstructure of annealed alloys, 200x.  
a - initial aluminum-titanium  $Ti_3Al$ ; b - alloyed 1%  
technical atmosphere-% V; c - the same, 1 technical  
atmosphere-% Mo; d - the same, 1% technical atmos-  
phere-% Fe; e - the same, 1 technical atmosphere-% Sn.

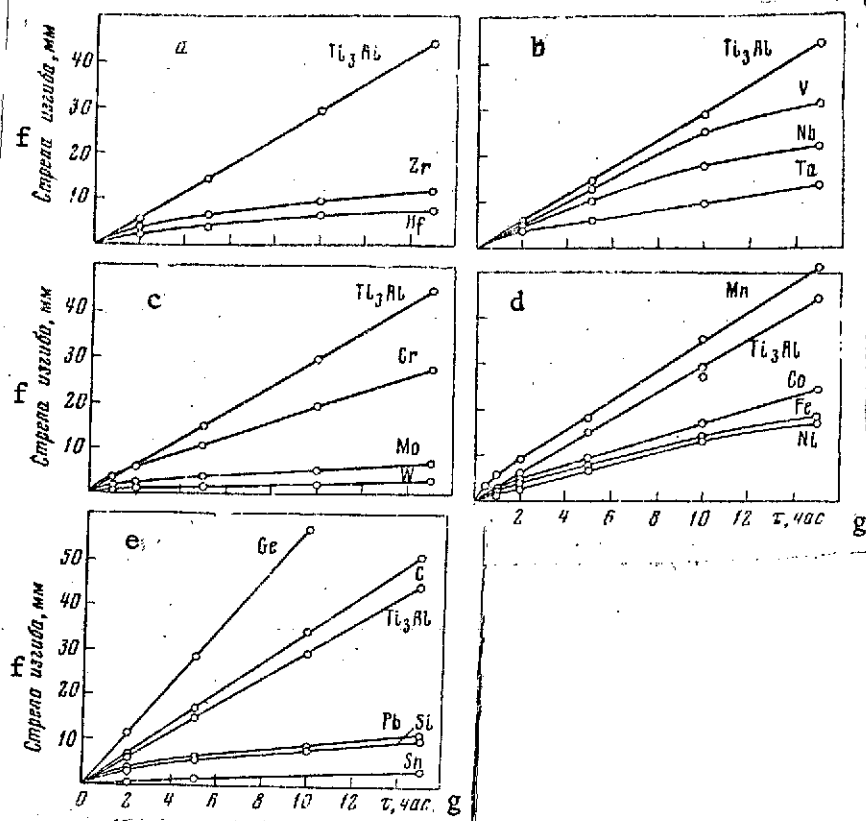


Fig. 2. Creep curves at  $750^{\circ}$  and  $\sigma = 25 \text{ kgf/mm}^2$  of aluminum-titanium  $\text{Ti}_3\text{Al}$ , alloyed at 1 technical atmosphere-%: Zr, Hf (a); Ta, Nb, V (b); W, Mo, Cr (c); Ni, Fe, Co, Mn (d); and Sn, Si, Pb, C, Ge (e).

Key: f. Bending indicator, mm  
g. Hours

The microstructure of aluminum-titanium with 1 technical atmosphere-% V or Nb, Ta, Cr is also single-phase, polyhedral (Fig. 1, b), and for an alloy with 1 technical atmosphere-% Mo (Fig. 1, c), it is two-phase. The change in the heat resistance of the  $\text{Ti}_3\text{Al}$  compound, depending on 1 technical atmosphere-% of the alloyed component V, Nb, Ta, Cr, Mo and W is shown in the form of creep curves in Fig. 2, b, c. The creep curves show that aluminum-titanium  $\text{Ti}_3\text{Al}$  is more heat resistant when alloyed with tantalum than niobium, and vanadium (Fig. 2, b). Of the metals of the VI group which increase the heat resistance are

tungsten, then molybdenum, and finally chromium (Fig. 2, c). These laws of the effect of alloying in the heat resistance of alloys are confirmed by the time taken for the bending indicator to show 10 mm. The bending indicator reached 10 mm for the  $Ti_3Al$  compound, as was said earlier, in 3 hours; an alloy with vanadium, 4 hours, with niobium, in 5 hours, with tantalum, in 10 hours; alloys with metal from the VI group are: with chromium, 4 hours, with molybdenum, 25 hours, with tungsten, 100 hours of deformation.<sup>1</sup> /103

The microstructure of the  $Ti_3Al$  alloy with 1 technical atmosphere-% Mn or Fe (Fig. 1, d) is polyhedral, and alloys with cobalt and nickel have a two-phase microstructure. The creep curves of these alloys are shown in Fig. 2, d. From them, it follows that from this series of alloys the most heat resistant are aluminum-titanium alloyed with Ni, then Fe and Co. Alloying with manganese also increases the creep rate of the  $Ti_3Al$  compound. The time taken for the bending indicator to reach 10 mm is: for the  $Ti_3Al$  compound, alloyed with manganese -- 2 hours, with nickel -- 7 hours, with iron -- 6 hours, and with cobalt -- 5 hours. When the  $Ti_3Al$  compound is alloyed with elements of the VIII group, the greatest heat resistance was shown by the alloy alloyed with nickel.

The microstructure of an alloy containing 1 technical atmosphere-% Si or Ge, Sn, Pb, is polyhedral (Fig. 1, e), the microstructure of an alloy with 1 technical atmosphere-% of C is two-phase. A comparison of the bending time of  $Ti_3Al$ , alloyed with the elements shown above of IVB group (Fig. 2, e), makes it possible to arrange the alloys in ascending values of heat resistance in the following order: aluminum-titanium, alloyed with tin, then silicon and lead. Carbon and germanium are

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<sup>1</sup>The time taken for the bending indicator to reach 10 mm for alloys with tungsten and tin was determined by extrapolation, assuming a uniform creep rate.

elements which do not increase the heat resistance of the  $Ti_3Al$  aluminum-titanium.

These uniformities are confirmed by the time taken for these alloys to reach the bending indicator of 10 mm. The time taken to reach the bending indicator at 10 mm by the  $Ti_3Al$  compound, alloyed with tin, was 65 hours; with silicon, 15 hours; with lead, 13 hours; with carbon, 3 hours; and with germanium, only approximately 2 hours.

### Conclusions

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1. The change of heat resistance of the  $Ti_3Al$  compound was investigated by the centrifugal bending method, depending on alloying it with one technical atmosphere-% of elements of IV-VIII and IVB groups of the periodic system: Zr, Hf, V, Nb, Ta, Cr, Mo, W, Mn, Fe, Co, Ni, C, Si, Ge, Sn, Pb. It was found that the heat resistance is greater when increasing the series number of the element inside each group.

2. Alloying  $Ti_3Al$  with vanadium or molybdenum considerably increases the deformability of alloys at high temperatures.

3. Alloys with an aluminum-titanium  $Ti_3Al$  base, alloyed with 1 technical atmosphere-% of W, Sn, Mo, Hf, Si, Zr, Ta, Nb, V have the greatest resistance to heat at  $750^\circ$  and a stress of  $25 \text{ kgf/mm}^2$ .



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